#### ARKIV FÖR KEMI Band 2 nr 37

Communicated 24 May 1950 by ARNE WESTGREN and PERCY QUENSEL

## Mixed oxides with layer lattices

### III. Structure of BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>

By BENGT AURIVILLIUS.

With 4 figures in the text

X ray studies on the compounds  $CaBi_2Nb_2O_9$  (1) and  $Bi_4Ti_3O_{12}$  (2) have shown that the comparatively complicated chemical formulae of these compounds can be explained by simple layer structures being built up from  $Bi_2O_2^{2+}$  layers and perowskite layers. The unit cells are pictured schematically in Figs. 1 a and 1 c. It was found both for  $CaBi_2Nb_2O_9$  and  $Bi_4Ti_3O_{12}$  that the symmetry was body-centered pseudo-tetragonal and that the length of the a axes had the same value (3.8 Å) while the length of the c-axis was 25 Å for  $CaBi_2Nb_2O_9$  and 33 Å for  $Bi_4Ti_3O_{12}$ . In both structures the heavy atoms form approximately a "substructure" with a smaller body-centered tetragonal cell with a = 3.8 Å and c = 25/5 Å for  $CaBi_2Nb_2O_9$  or c = 33/7 Å for  $Bi_4Ti_3O_{12}$ .

The  $\mathrm{Bi_2O_2}$  layers and perowskite layers lie perpendicular to the c-axis. Similar layer structures have been found for a number of bismuth oxicompounds (3,4,5). The common structural element in all these compounds is quadratic  $\mathrm{Bi_2O_2}$  layers between which halides or certain radicals are inserted. This explains the fact that the a axes of all these compounds are of about the same length. For a

survey see (5).

For the CaBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub> type each perowskite layer has the composition (CaNb<sub>2</sub>O<sub>7</sub>)<sub>n</sub> and the height of the layer is equal to four distances Nb—O or approximately to the height of two E2<sub>1</sub> (perowskite) unit cells (see Fig. 1 a). A compound with a somewhat similar structure has previously been investigated by LAGER-CRANTZ and SILLÉN (5). In this structure (see Fig. 1 b), beyerite CaBi<sub>2</sub>O<sub>2</sub>(CO<sub>3</sub>)<sub>2</sub>, the point positions corresponding to the positions of the Nb atoms in CaBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub> are occupied by "rotating" CO<sub>3</sub> groups.

For the Bi<sub>2</sub>Ti<sub>3</sub>O<sub>12</sub> type the perowskite layers have the composition (Bi<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub>)<sub>n</sub> and the height of the layer is equal to six distances Ti—O or approximately

to the height of three E21 unit cells.

The general formula for a compound built up in a way similar to CaBi<sub>2</sub>Nb<sub>2</sub>O<sub>3</sub> but where the height of the perowskite layer enclosed between a pair of Me<sub>2</sub>O<sub>2</sub> layers is equal to the height of m E2<sub>1</sub> cells, will be:

 $Me_2O_2$  ( $Me'_{m-1}R_mO_{8m+1}$ ). Me, Me': Ca, Sr, Ba Bi (K + Bi)/2 etc.

R: Ti, Nb, Ta (Nb + Ti)/2 etc.

### B. AURIVILLIUS, Mixed oxides with layer lattices. III

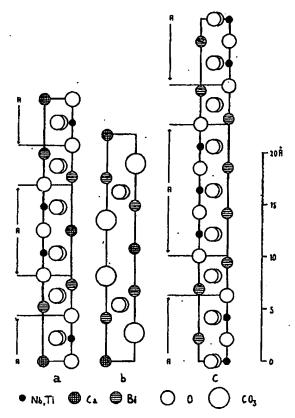


Fig. 1. Schematical pictures of the structures af a. CaBi<sub>2</sub>Nb<sub>2</sub>O<sub>2</sub> b. CaBi<sub>2</sub>O<sub>2</sub>(CO<sub>2</sub>)<sub>2</sub> and c. Bi<sub>4</sub>Ti<sub>2</sub>O<sub>12</sub>. The vertical lines indicate the lines 0 0 z and ½ ½ z in the unit cells. A denotes perowskitic (E2<sub>1</sub>) regions in the structures.

It seemed of interest to investigate whether compounds could be synthesized with m = 4. The present investigation shows that structures of the above type with m = 4 exist.

Mixtures of  $\mathrm{Bi_2O_3}$ ,  $\mathrm{BaCO_3}$  and  $\mathrm{TiO_2}$ , corresponding to the composition  $\mathrm{BaBi_4Ti_4O_{15}}$  were prepared and heated to  $1100^\circ\mathrm{C}$ . Single crystals, thin plates, were picked out and Weissenberg photographs were taken. These could be interpreted by means of a body-centered tetragonal cell with a=3.86 Å and c=41.7 Å. The strong lines of the powder photographs (taken from preparations heated to  $900^\circ\mathrm{C}$  (Au crucible) or  $1100^\circ\mathrm{C}$  (Pt crucible) were easily identified since they could all be described with the aid of the "sub-lattice" (a=3.86 c=41.7/9). If the c axis were 9 fold even the weak lines could be explained (Table 3). In this way the cell edges were found to be a=3.864 Å and c=41.76 Å. The observed density was 7.2, which agrees fairly well with the assumption of 2 formula units/unit cell (calculated density 7.49).

The Weissenberg photographs registered 0 k l, 1 k l, h h l and h, h + 1, l. In the Weissenberg and powder photographs there was nothing to indicate a lower Laue symmetry than  $D_4 - 4/mmm$ . Except for the extinctions due to the bodycentering, h k l occurring only for h + k + l = 2n, no systematic extinctions were found, which is characteristic of the space groups  $C_{4v}^0$ ,  $D_{2d}^{0,11}$  and  $D_{4d}^{17}$ . Fig. 2

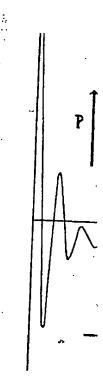


Fig. 2. Patterson



Fig. 3. Three dimmaterial was used the amplitudes are The ver

ARKIV FÖR KEMI. Bd 2 nr 37

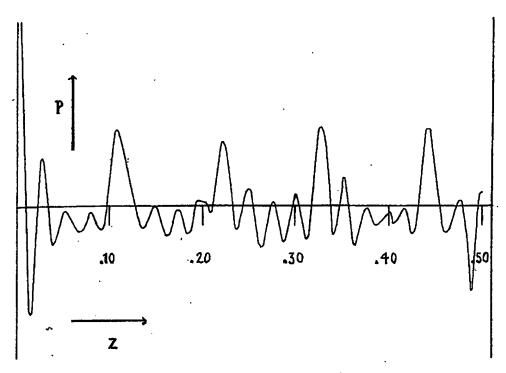


Fig. 2. Patterson Harker function of BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub> along 00 z. All observed reflexions were used for this calculation.

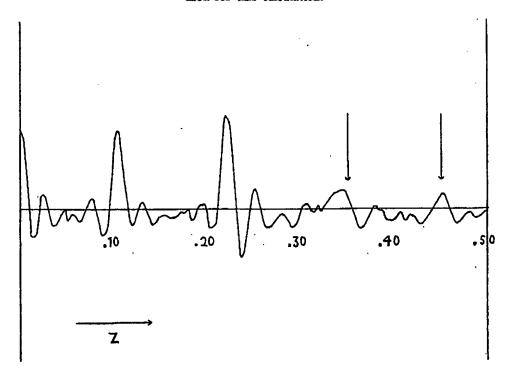


Fig. 3. Three dimensional Fourier cut along 0.0z for BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>. The same intensity material was used as for the Patterson Harker analysis pictured in Fig. 2. The signs of the amplitudes are the same as those obtained in the structure factor calculation for Table 2. The vertical arrows correspond to the  $z_{\rm Ti}$  values actually assumed.

O<sub>3</sub>)<sub>2</sub> and 4 denotes

thesized ve type

position
1 plates,
ould be
A and
preparaly iden" (a =
bould be
3.864 Å
ell with

i, l. In a lower ne bodyons were Fig. 2 B. AURIVILLIUS, Mixed oxides with layer lattices. III.

shows the Patterson Harker function along 0.0z. For all observed reflexions the  $F^2$  values were estimated from:  $F^2 \approx I_{\text{obs.}} \frac{\sin 2\theta}{1 + \cos^2 2\theta}$ . As expected large peaks appear at  $z \approx 1/9$  2/9 3/9 and 4/9.

It is interesting to compare the cell dimensions found above with the ones which might be expected if the compound  $BaBi_4Ti_4O_{15}$  is assumed to have a structure similar to that of  $CaBi_2Nb_2O_9$  but with m=4. In this case the a axis would have about the same value as found. The length of the c axis might be estimated in the following way: The c axis of  $Bi_3NbTiO_9$  (m=2 see (1)) is 25.11 Å; the value for  $Bi_4Ti_3O_{12}$  (m=3 see (2)) is 32.83, the difference is 7.72. If twice this value is added to the c axis of  $BaBi_2Nb_2O_9$  (m=2) the value 41.0 Å is obtained. The value actually found was, as mentioned above, 41.76 Å.

From the composition, cell dimensions, and crystal symmetry it seemed a priori probable that the structure of BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub> was the one we anticipated. Therefore, the parameters were worked out with the aid of the parameters found for

Table 1

Weissenberg photographs of  $BaBi_4Ti_4O_{15}$ . Cu  $K_\alpha$  radiation. For zero order photographs the regions of maximum absorption (see Wells (6)) are indicated by dotted lines. The intensities of 101, 103 and 105 have been taken from a zero order photograph, those of 1027—1049 from a first order photograph rotated around (100).

I.	001	201	221	401	111	311	3 3 1
			r1900			· <u> </u>	vw
E		_	vw		*****		vw
2 4 6 8 10					VW W	_	vvw
6	vw	VVW !	<b>VW</b>			vvw	vvw
8	m	· m	w		m		1
10	m	m.	W		st	vw.	w
12	w	vvw	vw		vvw		w
14	w	·	vvw_	_	w	_	
16 18	st	, w	m <sup>+</sup> .	vvw	m	AAM.	m
18	vst	m.	st	w	vst	w	st
20	st	) w	W	AAM	w		· - ·
22	m <sup>+</sup>	vvw	vvw	VVW	vw	_	vw
24	·w	vvw	vvw	vvw	vw		w
26	w	VVW			w	vvw	w
28	st	m <sup>+</sup>	₩+	m	m	vw	j
30	W		W	m+		vw	
32	w	w w+	l vor		m	w	
34	m	m	m <sup>+</sup>		_ m	m	
36	st	st	st	1	m+	· mo.	
36 38	vvw			, '	w	_	-
40	st	st	st		st	st	
42	vvw				vw		
44		_	ŀ	1			
46	m	m			st		1
48	i w	m	1		st	[	
50	₩*				m	l	ļ
52	₩+	1	<b>!</b>	•	_	]	
	1 "	<u> </u>	1	<u>'</u>	<del>!</del>	<u> </u>	·

Bi<sub>3</sub>NbTiC to be: 2 The pc 4 Ti in :  $\frac{1}{2}0(\frac{1}{2}-z_3)$ 4 O in  $\pm z_1 \approx 1/9$ 

These at same values ones only i be an incorr

The par of the Ba (000 + 00)Agreement  $\pm 0.004$ ,  $z_4$  lexions

l large

the a the a c axis see (1)) ence is above,

a priori Thereund for

· photodotted :o order

331

VW

vw

ARKIV FÖR KEMI. Bd 2 nr 37

Table 1 (cont.)

Weissenberg photographs of BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>. Cu K<sub>a</sub> radiation.

ı	101	211	301	321	411
,		w	vw	vw	_
1 1	m		\ <u>\ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ </u>		
1 3 5	vvw	VVW		7VW	
1 5	VVW	vw	<b>vvw</b>	w	
7	w	V W			700
9	vst	st —	m	m.	<b>w</b>
11		_			
13	w_	-			_
15	w <sup>+</sup>		VVW		
17	m	VVW	vvw		
19	st	vw	VVW	vvw.	VVW
21	w ·		vvw	VAM.	vvw
23	₩	_	vvw	vvw	m_
25	m <sup>+</sup>	vw	VW	VW.	m <sup>+</sup>
27	st	m	W	m.	
29	m	vw	vvw	vvw	
31	m.	w	vw	vw	
33	vw	<b>VVW</b>	vvw	vw	
35			· —	ļ	
37	$\mathbf{m}^{+}$	m	m		
39	vw	w	m	ļ	
41	w	w	w <sup>+</sup>	1	
43	₩ <sup>+</sup>	m		ļ	•
45	m	$\mathbf{m}^+$			
47	<u></u>				
49	st				
51				ļ	
	<u> </u>		1		

Bi<sub>3</sub>NbTiO<sub>9</sub> and Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub>. The positions of the heavy atoms might be expected to be: 2 Ba in 000, 4 Bi in  $\pm 00z_1$ , 4 Bi in  $\pm 00z_2$ .

The positions of the Ti and O atoms might be expected to be:

4 Ti in  $\pm 0.0z_3$ , 4 Ti in  $\pm 0.0z_4$ , 2 O in  $0.0\frac{1}{2}$ , 8 O in  $\pm (0\frac{1}{2}(\frac{1}{2}-z_3), \frac{1}{2}0(\frac{1}{2}-z_3))$ , 8 O in  $\pm (0\frac{1}{2}(\frac{1}{2}-z_4), \frac{1}{2}0(\frac{1}{2}-z_4))$ , 4 O in  $\pm 0.0(z_3+z_4)/2$ 4 O in  $\pm 0.0(z_3-(z_4-z_3)/2)$ , 4 O in  $0\frac{1}{2}\frac{1}{4}, \frac{1}{2}0\frac{1}{4}$  $z_1 \approx 1/9$   $z_2 \approx 2/9$   $z_3 \approx 0.350$   $z_4 \approx 0.450$ 

These atomic positions would give rise to high peaks in the Patterson-Harker plot at the same values as actually found. The calculated area ratios agree, however, with the observed ones only in as much as the biggest area is found for the peak at 1/9. The reasons might be an incorrect choice of the zero level and errors in the estimation of the intensities.

The parameters were varied around the above values for different positions of the Ba atoms: 2 Ba in 000,  $00z_1$  or  $00z_3$ , 2 Ba equally distributed over  $(000+00z_1)$ ,  $(000+00z_2)$ ,  $(00z_1+00z_3)$  or  $(000+00z_1+00z_2)$ . The best agreement seemed to be for  $z_1=0.106\pm0.001$ ,  $z_3=0.221\pm0.001$ ,  $z_3=0.352\pm0.004$ ,  $z_4=0.452\pm0.004$  with 2 Ba equally distributed over  $(000+00z_1+00z_2)$ 

#### B. AURIVILLIUS, Mixed oxides with layer lattices. III

Table 2
Weissenberg photographs of BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>. Cu  $K_{\alpha}$  radiation  $I_{\rm calo.} = (2.5 \text{ F/f}_{\rm Bi})^{3}$ 

				Teste. — (2.0 Tright)									
ı	I <sub>obs.</sub> 101	Icalc. 101	Iobs. 2 1 1	Icalc. 2 1 1	Iobs. 3 0 1	Icalc. 3 0 1							
1	m	73	. w	90 7.3	vw	94 8.4							
3	VVW	5.8 4.8	VVW	5.8	_	5.8							
5 7	vvw w	30	vw	31	vvw	32							
9	vst	620	st	610	m	630							
11		13	· —	12		14							
13	w	25	<b> </b>	24	l — i	24							
15	w <sup>+</sup>	62		60	vvw	60							
17	m	94 .	vvw	92	VVW	90							
19	st	160	vw	170	vvw	170							
21	w .	15		12	VVW	12 18							
23	w m <sup>+</sup>	19		18 120	VVW VW	120							
25 27		120 320	w m	320	w	320							
	st		1	1		44							
29	m	45 180	vw	45 160	VW VW	160							
31 33	m	160 <sup>-</sup> 56	w vvw	56	VVW	56							
35 ·	vw	13	VV#	12	<u>'''</u>	12							
37	m <sup>+</sup>	300	m	300	m.	300							
39	vw	12	w	13		13							
41		15	w	14	m w <sup>+</sup>	15							
43	w w <sup>+</sup>	200	m m⁺	200									
45	m	180	m <sup>+</sup>	180									
47		18		,									
49	st	250				•							
51		3.6	<del> </del>		<u></u>								
ı	Iobs. 0 0 1	Icalc. 0 0 1	<i>I</i> obs. 111	Icalc. 111	Iobs. 2 2 1	Icale. 2 2 1							
2				0.5	vw	0.6							
4			· vw	4.8		5.3							
6	vw ·	21	w	23	vw	24							
8	m.	110	m.	100	₩	110							
10													
	m	62	st .	280	w	98							
12	w	12	∇V₩	6.3	w · vw	98 9.6							
14	w w	12 14	∨vw w	6.3 16	W VVW	98 9.6 14							
14 16	w w st	12 14 83	vvw w m	6.3 16 79	w vw vvw m <sup>+</sup>	98 9.6 14 81							
14 16 18	w w st vst	12 14 83 440	vvw w m vst	6.3 16 79 480	w vw vvw m <sup>+</sup> st	98 9.6 14 81 450							
14 16 18 20	w w st vst st	12 14 83 440 170	vvw w m vat w	6.3 16 79 480 36	w vw vvw m <sup>+</sup> st w	98 9.6 14 81							
14 16 18 20 22	w w st vst st m	12 14 83 440 170 100	vvw w m vst	6.3 16 79 480 36 79 86	w vw vvw m <sup>+</sup> st	98 9.6 14 81 450 120							
14 16 18 20 22 24	w w st vst st	12 14 83 440 170 100	VVW W m vat W	6.3 16 79 480 36 79 86	w vw vvw m <sup>+</sup> st w vvw	98 9.6 14 81 450 120 92 78							
14 16 18 20 22 24 26	w w st vst st m+ w	12 14 83 440 170 100 81	vvw m vst w vw vw	6.3 16 79 480 36 79 86 36	w vw vvw m <sup>+</sup> st w vvw	98 9.6 14 81 450 120 92							
14 16 18 20 22 24 26	w w st vst st m w w	12 14 83 440 170 100	VVW W m vat W	6.3 16 79 480 36 79 86 36 240	w vw vvw vvw w w	98 9.6 14 81 450 120 92 78 39 280 1.4							
14 16 18 20 22 24 26	w w st vst st m+ w	12 14 83 440 170 100 81 39 280 2.6	vvw w m vst w vw vw m m	6.3 16 79 480 36 79 86 36 240 11 20	w vw vvw vvw w w	98 9.6 14 81 450 120 92 78 39 280 1.4 12							
14 16 18 20 22 24 26 28 30 32	w w st vst st m w w st	12 14 83 440 170 100 81 39 280 2.6 11	vvw w m vst w vw vw m m	6.3 16 79 480 36 79 86 36 240 11 20	w vw vvw — w w m+	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140							
14 16 18 20 22 24 26 28 30 32 34	w w st vst st m w st w st w	12 14 83 440 170 100 81 39 280 2.6 11 140 240	vvw w vvst vw vw m m m m	6.3 16 79 480 36 79 86 36 240 11 20 110 240	w vw vvw vvw w w	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140 240							
14 16 18 20 22 24 26 28 30 32 34 36 38	w w st vst st m w st w st vw	12 14 83 440 170 100 81 39 280 2.6 11 140 240	vvw w m vst w vw vw m m m	6.3 16 79 480 36 79 86 36 240 11 20 110 240 35	w vw vvw w w w st	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140 240 21							
14 16 18 20 22 24 26 28 30 32 34 36 38 40	w w st vst st m w st w st vvw st	12 14 83 440 170 100 81 39 280 2.6 11 140 240 19 380	vvw w w vst vw vw m m m m m w st	6.3 16 79 480 36 79 86 36 240 11 20 110 240 35 280	w vw vvw — w w m+	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140 240							
14 16 18 20 22 24 26 28 30 32 34 36 88 40 42	w w st vst st m w st w st vw	12 14 83 440 170 100 81 39 280 2.6 11 140 240 19 380 11	vvw w m vst w vw vw m m m	6.3 16 79 480 36 79 86 36 240 11 20 110 240 35 280 22	w vw vvw w w w st	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140 240 21							
14 16 18 20 22 24 26 28 30 32 34 36 38 40 42 44	w w st vst st m w st w st vv st vv st vv st	12 14 83 440 170 100 81 39 280 2.6 11 140 240 19 380 11	vvw w vst vw vw m m m m m w st vw	6.3 16 79 480 36 79 86 36 240 11 20 110 240 240 25 280 22 4.8	w vw vvw w w w st	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140 240 21							
14 16 18 20 22 24 26 28 30 32 34 36 38 40 42 44 46	w w st vst m w st w vv st vv n t	12 14 83 440 170 100 81 39 280 2.6 11 140 240 19 380 11 3.2 240	vvw w vst vw vw m m m m m t vv st	6.3 16 79 480 36 79 86 36 240 11 20 110 240 35 280 22 4.8 230	w vw vvw w w w st	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140 240 21							
14 16 18 20 22 24 26 28 30 32 34 36 38 40 42 44	w w st vst st m w st w st vv st vv st vv st	12 14 83 440 170 100 81 39 280 2.6 11 140 240 19 380 11	vvw w vst vw vw m m m m m w st vw	6.3 16 79 480 36 79 86 36 240 11 20 110 240 240 25 280 22 4.8	w vw vvw w w w st	98 9.6 14 81 450 120 92 78 39 280 1.4 12 140 240 21							

: -
hkl
0010 101 105
1 0 9 1 1 0 0 0 16
11 6 11 8 0018
1 1 10 1 0 15 0 0 20
1 0 17 2 0 0 1 0 19
1 1 16 2 0 8 {1 1 18
1 0 21 2 0 10 2 1 1
\begin{pmatrix} \{2 & 1 & 1 \\ 2 & 1 & 3 \\ \{2 & 1 & 7 \\ \{1 & 1 & 20 \\ \{2 & 0 & 1 & 1 \\ \{2 & 0 & 1 \\ \$1
{2 0 14 2 1 9

 $00z_2$ ). Oth seen the thr following M over the po From Tal  $\approx I_{22l} \approx I_4$  be expected 00z,  $\frac{1}{2}$   $\frac{1}{2}z$ , for the rows senberg phothe observed more clearly ≥ 3039. Th the z parame structure fac are 222:22. reflexions the be introduced the agreemen

calc.

0.6

5.3

:80 1.4

.40

:40

9.8  ARKIV FÖR KEMI. Bd 2 nr 37

Table 3

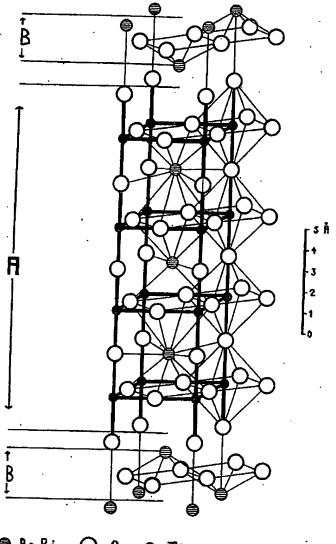
Powder photographs of BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub> Cr  $K_{\alpha}$  radiation.

آ	h k l	$10^4$ $\sin^2 \theta$ calc.	$10^4$ $\sin^2 \theta$ obs.	Iobs.	hkl	10 <sup>4</sup> sin <sup>2</sup> θ calc.	I0 <sup>4</sup> sin <sup>2</sup> θobs.	Iobs.
-						1		
1	0010	753	745	w	2016	5443	5440	VW
ı	10 1	887	878	m	1025	5582	5581	w
ı	10 5	1067	1058	vw	2113	5667	5669	∇VW
1	10 9	1489	1481	st	0028	5900	5892	vw
Ĺ	11 0	1758	1751	m	2018	5954	5960	m.
ı	0016	1927	1915	vvw	<b>∫2115</b>	6088	6089	
1	11 6	2031	2018	vw	11124	6093	0088	VW
١	11 8	2240	2236	vw	1027	6365	6364	m
-			2423.	m_	2 0 20	6526	6529	vw broad
1	0018	2438 2511	2504	m.+	(220	7032		1
	1 1 10		2567 ·	vvw	222	7064	7070	w broad
L	1 0 15	2572	1 -	VVW W	2 1 19	7112		
i	0 0 20	3010	2992		130 5	8099		
ı	1017	3054	3051	W	1031	8111	8104	VVW
l	200	3516	3514	w <sup>+</sup>	2214	8114	0	
1	1 0 19	3596	3600		2 1 23	8376	8365	VW
Ì	1 1 16	3685	3681	w	(2214	8507	0000	
	208	3998	3996	vw	30 9	8521	8520	m
1	<b>∫1 1 18</b>	4196	4186	m	11130	8531	0020	]
l	1021	4198	ł		310	8802		
	2010	4269	4265	w	3011	8812	8802	m.
1	<i>[</i> 211	4403	4402	vw	31 2	8820	0002	
	1213	4418	1	,	1033	9074		_
ı	∫21 7	4764	4774	vw	2 1 25	9098	9092	w broad
1	11 1 20	4768	±.,±	'"				1
l	. \$2014	4991	5009	m <sup>+.</sup>	11132	9464	9484	w broad
ı	121 9	5005	1.		{2 2 18	9470	[	Į.

 $00z_2$ ). Other distributions of Ba should, however, not be excluded; as will be seen the three dimensional Fourier cut along 00z (see Fig. 3) seems to favour the following Me arrangement:  $4 \text{ Bi } \pm 00z_2$  and (2 Ba + 4 Bi) equally distributed over the positions 000 and  $\pm 00z_1$ .

From Table 1 it is seen that roughly for the same value of l:  $I_{00l} \approx I_{20l} \approx I_{22l} \approx I_{40l}$ ,  $I_{11l} \approx I_{31l} \approx I_{33l}$ ,  $I_{10l} \approx I_{21l} \approx I_{30l} \approx I_{32l} \approx I_{41l}$  as might be expected from the above atomic positions with atoms only on the lines 00z,  $\frac{1}{2}$ ,  $\frac{1}{2}z$ , 0,  $\frac{1}{2}z$  and  $\frac{1}{2}0z$ . Table 2 gives calculated and observed intensities for the rows 00l, 10l, 11l, 21l, 22l and 30l. It was found from the Weissenberg photographs that 0028 > 0030 and 1037 > 1039 (see Table 2) but the observed ratios do not seem to be as large as those calculated. This is more clearly seen for the rows 40l and 30l where 4028 < 4030 and  $3037 \ge 3039$ . These discrepancies could neither be removed by small variations in the z parameters nor by assuming other discrepancies found from Tables 1 and 2 are 222:224, 332:334, 0050:0052 and 1148:1152. However, for most reflexions the agreement is quite good and considering the errors which might be introduced by absorption effects and errors in the ratios  $f_{Tl}/f_{Bl}$  and  $f_0/f_{Bl}$  the agreement might on the whole be classified as fairly good.

# B. AURIVILLIUS, Mixed oxides with layer lattices. III



● Ba Bi O 0 • Ti

Fig. 4. One half of the unit cell of BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>. A denotes the perowskitic region and B the Me<sub>2</sub>O<sub>2</sub> layers.

The following structure is therefore proposed:  $D_{44}^{17} - I4/mmm$  (000,  $\frac{1}{2}\frac{1}{2}\frac{1}{2}$ ) +

2 Me<sub>1</sub> in 2 (a) 000 2 O<sub>1</sub> in 2 (b) 00 $\frac{1}{2}$ 8 O<sub>2</sub> in 8 (g) 0 $\frac{1}{2}z$ ; 0 $\frac{1}{2}\bar{z}$ ;  $\frac{1}{2}$ 0z;  $\frac{1}{2}$ 0 $\bar{z}$  z = 0.048 4 Ti<sub>1</sub> in 4 (e) 00z; 00 $\bar{z}$  z = 0.452 4 O<sub>3</sub> in 4 (e) z = 0.402

526

## BEST AVAILABLE COPY

Ba aı

If 1 the st ½ ½ 0; ½ ½ z; 8 (i) z 8 Me<sub>2</sub>

The

 $0_1 - \{$   $0_3 - \{$  $0_3 - \{$ 

05-4

One h:

I wi

Stoc.

REF. 499. — G., Dis. N:o 20,

ARKIV FÖR KEMI. Bd 2 nr 37

4 Me<sub>2</sub> in 4 (e) 
$$z = 0.106$$
  
8 O<sub>4</sub> in 8 (g)  $z = 0.148$   
4 Ti<sub>2</sub> in 4 (e)  $z = 0.352$   
4 O<sub>5</sub> in 4 (e)  $z = 0.302$   
4 Me<sub>3</sub> in 4 (e)  $z = 0.221$   
4 O<sub>6</sub> in 4 (d) O ½ ½; ½ O ½

Ba and Bi equally distributed over all Me positions.

If the structure is described by means of an orthorhombic space group  $(D_{2k}^{23})$ , as used for the structures of  $Bi_3NbTiO_9$  and  $Bi_4Ti_3O_{12}$ , the positions will be:  $D_{2k}^{23} - F \ m \ m \ m \ (0.00; \frac{1}{2}\frac{1}{2}0; 0\frac{1}{2}\frac{1}{2}; \frac{1}{2}0\frac{1}{3}) + 4 \ Me_1 \ \text{in} \ 4 \ (a) \ 0.00, \ 4 \ O_1 \ \text{in} \ 4 \ (b) \ 0.0\frac{1}{2}, \ 16 \ O_2 \ \text{in} \ 16 \ (j) \ \frac{1}{4}\frac{1}{4}z; \ \frac{1}{4}\frac{1}{2}\overline{z}; \frac{1}{4}\frac{1}{2}\overline{z}; z = 0.048, \ 8 \ Ti_1 \ \text{in} \ 8 \ (i) \ 0.0z; \ 0.0\overline{z} \ z = 0.452, \ 8 \ O_3 \ \text{in} \ 8 \ (i) \ z = 0.402, \ 8 \ Me_3 \ \text{in} \ 8 \ (i) \ z = 0.106, \ 16 \ O_4 \ \text{in} \ 16 \ (j) \ z = 0.148, \ 8 \ Ti_2 \ \text{in} \ 8 \ (i) \ z = 0.352, \ 8 \ O_5 \ \text{in} \ 8 \ (i) \ z = 0.302, \ 8 \ Me_2 \ \text{in} \ 8 \ (i) \ z = 0.221, \ 8 \ O_6 \ \text{in} \ 8 \ (f) \ \frac{1}{4}\frac{1}{4}\frac{1}{4}; \ \frac{1}{4}\frac{1}{4}\frac{8}{4}.$ 

The distances (A) and coordination will be:

One half of the unit cell is shown in Fig. 4.

I wish to thank Professor L. G. Sillen for valuable discussions on this work.

Stockholms Högskola, Institute of Inorganic and Physical Chemistry. May 1950.

REFERENCES. 1. Aurivillius, B., Arkiv Kemi 1 (1950) 463. — 2. —, Ibid, 1 (1950) 499. — 3. Bannister, F. A. and Hey, M. H., Miner. Mag. 24. (1935) 49. — 4. Sillén, L. G., Dissert., Stockholm 1940. — 5. Lagercrantz, Å. and Sillén, L. G., Arkiv Kemi 25 N:o 20, 1948. — 6. Wells, A. F., Z. Krist. 96 (1937) 451.

Tryckt den 14 oktober 1950

Uppsala 1950. Almqvist & Wiksells Boktryckeri AB

527

n and